Simultaneous depth profiling of the $^{12}\text{C}$ and $^{13}\text{C}$ elements in different samples using (d,p) reactions
Colaux, Julien; Terwagne, Guy

Published in:
Nuclear instruments and methods in physics research B

Publication date:
2005

Document Version
Early version, also known as pre-print

Link to publication
Citation for published version (HARVARD):
Colaux, J & Terwagne, G 2005, 'Simultaneous depth profiling of the $^{12}\text{C}$ and $^{13}\text{C}$ elements in different samples using (d,p) reactions', Nuclear instruments and methods in physics research B, vol. 240, no. 1-2, pp. 429-433.
Simultaneous depth profiling of the $^{12}$C and $^{13}$C elements in different samples using (d,p) reactions

Julien L. Colaux*, Guy Terwagne

Laboratoire d'Analyse par Réactions Nucléaires, Facultés Universitaires Notre-Dame de la Paix, 61, rue de Bruxelles, B-5000 Namur, Belgium

Available online 2 August 2005

Abstract

Nuclear reactions (d,p) are often used to perform depth profiling of light elements in solids. In particular, protons coming from $^{12}$C(d,p)$^{13}$C and $^{13}$C(d,p)$^{14}$C reactions are emitted at very different energies. Consequently these two reactions can be used to depth profile $^{12}$C and $^{13}$C simultaneously. Nevertheless the cross-section of $^{13}$C(d,p)$^{14}$C reaction is 10 times smaller than the $^{12}$C(d,p)$^{13}$C one. So, the geometry of detection must be judiciously chosen in order to depth profile these two elements with a high sensitivity and good resolution.

In the framework of this study we have performed 400 keV $^{13}$C ions implantation into polished copper substrates at different temperatures and implanted doses with a 2 MV Tandem accelerator. Using the reactions described above, we have studied the evolution of $^{13}$C depth profile as a function of implanted doses and temperature. We have also determined the origin of surface contamination that appears during the implantation process.

© 2005 Elsevier B.V. All rights reserved.

PACS: 25.45.De; 29.30.Ep

Keywords: $^{12}$C; $^{13}$C; (d,p); Profile concentration; NRA

1. Introduction

Since 1985 and the fullerene ($C_{60}$) discovery [1], there has been an intensive effort driven by the scientific community to study the so-called novel carbon nanostructures (fullerenes, nanotubes, carbon onions). At present there are numerous techniques allowing to synthesize carbon onions [2–5]. One of them developed by Cabioc'h et al. in the early 1990s uses high dose carbon ion implantation into hot substrates (like copper or silver) [4,6–9]. Recently the same team has been shown by successive $^{13}$C and $^{12}$C low energy implantation into silver that depth distribution of implanted carbon has
a Gaussian-like shape and there is no preferential carbon diffusion towards the surface during the implantation process [10].

In this paper, we study high energy implantation of 13C into copper and we try to explain the depth distributions observed as a function of implanted dose and temperature. On the other hand, it is well known that build-up of carbon occurs during ion implantation. This phenomenon is directly related to residual vacuum and implantation parameters and we try to show that it is responsible to the surface contamination observed on the implanted samples. We optimized the detector geometry in order to depth profile 12C and 13C with high sensitivity and good resolution by using nuclear reactions induced by deuteron.

2. Experimental details

Polished copper substrates were fixed into a furnace mounted in an implantation chamber to perform different 13C+ ions implantations at different temperatures (200, 370 and 600 °C). Each one was performed in residual pressure less than 7 × 10^−4 Pa. 13C+ ions of 400 keV were produced by ALTAIS$^1$ the 2 MV Tandetron accelerator installed at LARN (Namur). About 4 h of implantation was necessary to obtain an average dose of 5 × 10^{17} at. cm^−2 on a rectangular surface (5 × 6 mm^2) with ion beam intensity of about 1.6 μA. Due to the fact that the beam was not homogenous, we are able to measure the areal density at different places on the implanted specimens using the reactions described below.

In order to characterize the implanted samples in one measurement, we used non-resonant nuclear reactions (NRA) induced by 1.050 MeV deuteron beam (1 × 1 mm^2). Two silicon surface barrier detectors were mounted at 135° and 165° relative to the incident beam. The NRA detector at 135° measured protons emitted by 12C(d,p0)13C and 13C(d,p0)14C reactions. A 12 μm Mylar™ absorber foil was placed in front of the detector to stop the intense flow of backscattered ions. Since the cross-section of the 13C(d,p0)14C is 10 times lower (about 3 mb sr^−1) [11] than the cross-section of 12C(d,p0)13C [12–14], we have chosen to use a large solid angle (25.7 msr) in order to obtain a good statistic while limiting the acquisition time to about 10 min. The RBS detector placed at 165° was collimated with a very small solid angle (0.18 msr) in order to detect backscattered deuterons flow without any absorber. The backscattered deuterons on copper were used as a monitor to measure incident particles [15]. Each of the three samples was measured in five different places in order to evaluate the non-uniformity of carbon beam irradiation. The acquisition time for each measurement was monitored by integrating the current beam to obtain an accumulated charge of 15 μC.

![Experimental NRA spectra of copper samples implanted with 13C+ at various temperatures (T) and fluences (d) recorded in the NRA detector. Symbols represent the experimental spectra and the black lines represent simulations realized with SIMNRA code.](image-url)
3. Results

The measurements were realized at different places on each sample and show that the implanted dose varied in a wide range (from $2 \times 10^{17}$ to $16.3 \times 10^{17}$ at. cm$^{-2}$) according to the position on the irradiated zone. These doses were calculated by simulation of the experimental spectra with the SIMNRA code [16]. As indicated in the previous section, the average implanted dose is $5 \times 10^{17}$ at. cm$^{-2}$ and we have reactions with a good yield to reveal the non-uniformity which is not the first objective of this paper.

Fig. 1 shows typical experimental NRA spectra of copper samples implanted with $^{13}$C at various temperatures recorded in the NRA detector. The $^{12}$C$_{p0}$ at 2.9 MeV and the $^{13}$C$_{p0}$ at 5.9 MeV are due to nuclear reactions $^{12}$C(d,p$_0$)$^{13}$C and $^{13}$C(d,p$_0$)$^{14}$C respectively. We have adopted the same identification for the reactions induced by deuterons (i.e. $^{16}$O$_{p0}$ as nuclear reaction $^{16}$O(d,p$_0$)$^{17}$O). We can observe that the intensity of the $^{13}$C$_{p0}$ peak is proportional to the implanted dose. The shape of this peak is wider for the sample implanted at high temperature (Fig. 1(a)). This effect due to diffusion phenomenon during implantation will be discussed later.

The intensity of the $^{12}$C$_{p0}$ peak reveals the build-up phenomenon during the implantation. At temperature lower than 370 °C (Fig. 1(b) and (c)) we can observe a very intense contribution of $^{12}$C at the surface while $^{12}$C contamination is strongly reduce when the implantation temperature is 600 °C (Fig. 1(a)). As it can be seen in Fig. 1, a shoulder at low energy of the $^{12}$C$_{p0}$ peak reveals the high sensitivity of this reaction. We can measure the areal density of $^{12}$C in the bulk of the implanted samples.

Table 1

<table>
<thead>
<tr>
<th>Implantation Temperature</th>
<th>Implanted Sample 13C (10$^{15}$ at. cm$^{-2}$)</th>
<th>Implanted Sample 12C$_{surf}$ (10$^{15}$ at. cm$^{-2}$)</th>
<th>Implanted Sample 16O (10$^{15}$ at. cm$^{-2}$)</th>
</tr>
</thead>
<tbody>
<tr>
<td>200 °C</td>
<td>$6.1 \times 10^{2}$</td>
<td>$2.75 \times 10^{2}$</td>
<td>$6.9 \times 10^{1}$</td>
</tr>
<tr>
<td>370 °C</td>
<td>$15.4 \times 10^{2}$</td>
<td>$1.38 \times 10^{2}$</td>
<td>$5.8 \times 10^{1}$</td>
</tr>
<tr>
<td>600 °C</td>
<td>$16.3 \times 10^{2}$</td>
<td>$0.18 \times 10^{2}$</td>
<td>$2.0 \times 10^{1}$</td>
</tr>
</tbody>
</table>

Fig. 2. Depth distributions of (a) $^{13}$C, (b) $^{12}$C and (c) $^{16}$O for 400 keV $^{13}$C$^+$ implantations into copper performed at different temperatures ($T$) and for various fluences ($d$).

Fig. 3. Depth distributions of $^{13}$C for 400 keV ion implantation into copper performed at 370 °C with various fluences ($d$).
At low energy (<2.5 MeV) the spectra for the high temperature (600°C) and the low temperatures (200°C and 370°C) are completely different. We can observe in Fig. 1(a) the peaks $^{16}\text{O}_0$ and $^{16}\text{O}_1$, due to the presence of oxygen, at 2.2 and 1.35 MeV respectively. The sensitivity of these reactions is limited by an interference with the $^{13}\text{C}(d, a_0)^{11}\text{B}$. In order to simulate the spectra at low energies, the cross-section of this reaction already measured by Marion et al. [17] for energies higher than 1 MeV should be measured at lower energies.

4. Discussion

Fig. 2 shows the depth distributions of $^{12}\text{C}$, $^{13}\text{C}$ and $^{16}\text{O}$. The concentration of samples has been sliced into layers containing all concerned elements, which are represented by the different points. The surface contribution ($^{12}\text{C}_{\text{surf}}$) is clearly observed in Fig. 2(b) and the areal density of $^{12}\text{C}$ has been calculated and reported in Table 1. This surface contribution is due to build-up of $^{12}\text{C}$ during the implantation. The sample implanted at 600°C contains the lowest $^{12}\text{C}$ contamination. This effect is explained by the dependence of the sputtering yield to the temperature. As the implantation temperature is increasing, the sputtering yield is also increasing [18,19].

At medium implantation temperature (370°C) the $^{13}\text{C}$ depth profiles show a Gaussian shape (Fig. 3). The projected range increases with the implanted dose. The experimental and theoretical projected ranges, calculated with SRIM2003 code [20], are reported in Table 2. The theoretical $R_p$ values have been calculated for a target composition corresponding to the half maximum of carbon contents (fourth column in Table 2). It is clearly seen in Table 2 that the agreement between experimental and calculated values is very good.

The $^{13}\text{C}$ depth profile at high temperature (600°C) shows a dissymmetrical distribution (Fig. 2(a)) which is correlated to the $^{12}\text{C}$ depth distribution (Fig. 2(b)). The wide distribution of $^{13}\text{C}$ is explained by diffusion process during implantation. This effect should be confirmed with a theoretical model like TRIDYN code [21] adapted to take into account the diffusion process. The correlation between $^{13}\text{C}$ and $^{12}\text{C}_{\text{bulk}}$ distribution is also explained by diffusion of $^{12}\text{C}$ during the implantation process.

Moreover, the oxygen depth distribution of the high implantation temperature is correlated to $^{13}\text{C}$ depth distribution. This effect should also be confirmed when $^{13}\text{C}(d,a_0)^{11}\text{B}$ will be measured at low energies.

5. Conclusion

We have developed an ideal geometry of detection in order to analyse simultaneously $^{12}\text{C}$, $^{13}\text{C}$ and $^{16}\text{O}$ by using nuclear reactions induced by deuterons. Two detectors were used to perform these measurements. The first detector positioned in RBS geometry was used as a monitor and the second one placed in NRA geometry showed a large solid angle to depth profile $^{12}\text{C}$, $^{13}\text{C}$ and $^{16}\text{O}$ simultaneously. Depth profiling of $^{12}\text{C}$ reveals a surface contamination due to $^{12}\text{C}$ build-up during implantation process and small bulk component which is correlated with $^{13}\text{C}$ depth distribution. The sensitivity of the $^{13}\text{C}(d,p_0)^{12}\text{C}$ nuclear reaction allows to detect 0.1 at.% of $^{12}\text{C}$. At low implantation temperature (<370°C), the $^{13}\text{C}$ depth profiles show a

<table>
<thead>
<tr>
<th>Dose (10(^{17}) at. cm(^{-2}))</th>
<th>$^{13}\text{C}_{\text{max}}$ (at.%)</th>
<th>($R_p$)(_{\text{exp}}) (10(^{15}) at. cm(^{-2}))</th>
<th>Composition of target</th>
<th>($R_p$)(_{\text{th}}) (10(^{15}) at. cm(^{-2}))</th>
</tr>
</thead>
<tbody>
<tr>
<td>15.4</td>
<td>40</td>
<td>4300</td>
<td>$^{0.200}\text{C}_{0.800}$</td>
<td>4140</td>
</tr>
<tr>
<td>6.8</td>
<td>23</td>
<td>4040</td>
<td>$^{0.115}\text{C}_{0.885}$</td>
<td>3930</td>
</tr>
<tr>
<td>5.8</td>
<td>17</td>
<td>4020</td>
<td>$^{0.083}\text{C}_{0.915}$</td>
<td>3950</td>
</tr>
<tr>
<td>5.0</td>
<td>15</td>
<td>3960</td>
<td>$^{0.073}\text{C}_{0.925}$</td>
<td>3880</td>
</tr>
<tr>
<td>2.0</td>
<td>6.5</td>
<td>3940</td>
<td>$^{0.032}\text{C}_{0.968}$</td>
<td>3810</td>
</tr>
</tbody>
</table>

The fourth column indicates the composition of the target used to calculate the ($R_p$)\(_{\text{th}}\) with the SRIM code.
Gaussian shape with \((R_p)_{\text{exp}}\) close to theoretical range \((R_p)_{\text{th}}\) calculated with SRIM code. For high temperature of implantation \((600 \, ^\circ\text{C})\), the \(^{13}\text{C}\) depth profile shows a diffusion process.

Finally we were able to make obvious that only first \(1000 \times 10^{15}\) at. cm\(^{-2}\) samples are weakly oxidized as long as implantation temperature is maintained below \(370 \, ^\circ\text{C}\). On the other hand oxidation is 10 times higher for implantation realized at \(600 \, ^\circ\text{C}\). Moreover \(^{16}\text{O}\) depth profile seems to be correlated to \(^{13}\text{C}\) depth distribution. In order to measure precisely oxygen, the cross-section of \(^{13}\text{C}(d,a)^{11}\text{B}\) nuclear reaction should be measured for projectiles energies below 1 MeV.

References